pected anomaly was found: the very broad and strong absorption lies in the region of 400–1600 cm⁻¹ with its maximum at 900 cm⁻¹.

Judging from the results so far available, which are consistent with other Type A acid salts, it seems probable that the $O(1)\cdots O(1^i)$ bond is not only crystallographically, but also truly, of the symmetrical singleminimum type. The problem of the symmetrical OHO bond is interesting, since a model of order-disorder type is accepted in most hydrogen-bonded ferroelectrics. Structure analysis in the ferroelectric phase (at about 80°K) is now in progress. A more detailed discussion will be given on the basis of the structures of the para- and ferroelectric phases.

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Barium Aluminate Hydrates. V.* The Crystal Structure of γ-Ba[AlO(OH)₂]₂

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 γ -BaO.Al₂O₃.2H₂O is orthorhombic, *Fdd*2, with a = 20.70, b = 8.977, c = 6.416 Å, Z = 8. X-ray structural analysis shows that it contains chains of composition $[AlO(OH)_2^{-1}]_{\infty}$, made up of tetrahedra sharing corners. Its formula should therefore be written γ -Ba $[AlO(OH)_2]_2$.

Introduction

Carlson & Wells (1948) were the first to produce monobarium aluminate dihydrate and their compound was designated ' α ' by Thilo & Gessner (1965) when a second polymorph (' β ') was found. Ahmed & Dent Glasser (1970) solved the structure of the α -polymorph and (1971) prepared a third compound of the same empirical formula which they called ' γ '. This present work deals with the γ modification. Crystal data from Ahmed & Dent Glasser (1971) are reproduced in Table 1. The compound crystallizes as plates lying on (100).

^{*} Part IV: Dent Glasser & Giovanoli, Acta Cryst. (1972). B28, 519.

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Table 1. Crystal data of y-BaO. Al₂O₃. 2H₂O

$= 20.70 \pm 0.02 \text{ Å}$
$= 8.977 \pm 0.004$
$= 6.416 \pm 0.003$
$= 3.25 \text{ g.cm}^{-3}$
$= 3.35 \text{ g.cm}^{-3}$

Unit cell content Ba₈Al₁₆O₄₈H₃₂

Experimental

Intensity data were collected using a Hilger and Watts Y190 automatic linear diffractometer with Mo $K\alpha$ radiation ($\lambda = 0.7107$ Å). The rotation axis of the crystal was c, and reflexions throughout one half of reciprocal space up to l=8 were measured. This gave a total of 1629 reflexions not systematically absent; agreement between equivalent reflexions was very good. After averaging equivalent reflexions a total of 529 reflexions remained, of which 71 were not considered to be significantly above background and were treated as 'unobserved'. Intensities were converted to structure factors in the usual way. No corrections were made for absorption or extinction. Data were processed on an ICL 4/50 computer, mainly using programs supplied by Dr F. R. Ahmed and collaborators of the National Research Council of Canada, and adapted for use on this machine by Mr J. S. Knowles of the Department of Computing, University of Aberdeen. Scattering factors for Ba^{2+} and Al^{3+} were taken from International Tables for X-ray Crystallography (1962). A curve for O²⁻ was constructed as described by Ahmed & Glasser (1970).

Structure determination

Approximate positions for the barium, aluminum and oxygen atoms were derived from the Patterson function and refined by the method of least squares (blockdiagonal approximation). Reflexions were weighted according to $W=1/\{1+[(|F_o|-P_2)/P_1]^2\}$. P₁ and P₂ were chosen initially to give maximum weight to reflexions of moderate intensity; subsequently an error analysis showed that the scheme selected was reasonable.

The Patterson function could be satisfied by two solutions. One set of parameters refined readily to a



Fig. 1. The structure of γ -Ba[AlO(OH)₂]₂ projected down c; heights are in c/100. Oxygen atoms are at the corners of the tetrahedra, and the heights of the corresponding aluminium atoms (small circles containing crosses) are shown in the upper half of the diagram. Large open circles indicate barium atoms. The upper half of the cell is drawn to emphasize the twofold axes on which the barium atoms lie; the lower emphasizes the perspective of the criss-crossing chains.

final R value of 0.055, while the other stuck at 0.16. Moreover, the first set gave a chemically satisfactory set of bond lengths and angles, while the second did not. The first solution is thus assumed to be the correct one. The corresponding parameters are given in Table 2, and the observed and calculated structure factors in Table 3. In addition to the 'unobserved' reflexions, 8

Table 2. Final parameters for y-Ba[AlO(OH)₂]₂

Figures in brackets represent the estimated standard deviation corresponding to the least significant digit.

-		Positional parameter		Isotropic temperature factor
	x/a	y/b	z/c	Biso
Ba	´ 0	0	0	1.05 (1)
Al	0.0860 (2)	0.3556 (3)	0.0519 (8)	0.46 (4)
O(1)	0.1307 (4)	0.194 (1)	0.017 (1)	0.3(1)
O(2) (hydroxyl)	0.0022(4)	0.305 (2)	0.114 (3)	$1 \cdot 1 (2)$
O(3) (hydroxyl)	0.0831 (7)	0.474 (1)	0.835 (2)	1.0 (2)

reflexions where the agreement between F_o and F_c was poor were omitted from the final cycles of refinement. These corresponded mainly either to very weak reflexions at high angles, or to reflexions very close to c^* which the mechanical construction of the Y190 makes it difficult to measure accurately.

Description of the structure

Fig. 1 shows the structure viewed along c. Each aluminum atom is surrounded tetrahedrally by four oxygen atoms. These tetrahedra share corners [O(1)] to form chains, and it is assumed that the two unshared corners [O(2) and O(3)] correspond to hydroxyl groups. The temperature factors (Table 2) are in accordance with this assumption.

The barium atoms lie on the twofold axes, and the chains lie along the *d*-glide planes perpendicular to **a**. This means that the chains run diagonally across the cell, with alternate chains running in opposite directions (Figs. 1 and 2). We believe that this is the first chain structure in which the chains do not run parallel to one another. There is no hydrogen bonding between the chains.

Details of the coordination polyhedra about Al and Ba are given in Table 4. The barium atom is surrounded by eight oxygen atoms which form a very distorted cube (Fig. 1). The four independent distances [Table



Fig. 2. The structure of γ -Ba[AlO(OH)₂]₂ viewed along **a**, showing how the anion chains criss-cross. For simplicity only one half the unit cell, from 0 to a/2, is shown. Heights of oxygen atoms are expressed in a/100.

Table 3. Observed and calculated F values ($ imes$ 1

Unobserved reflexions are marked with an asterisk.

1 20 20	L FO FC	L FO FC	L FO FC	L FO FC	L FO FC	L PD PC	L FO FC	L F0 FC	L FO F7		
H= 2. X= 0	5 839 988			3 801 802		H= 11, K= 5	4 397 391	1 2630 2540			
2 4174 4207	7 1010 1400	No 2, Ko 2	N# 22, K# 2	5 1040 1064	0 793 676	1 1071 1101	6 1631 1543	3 1930 1831	H# 6, E# 8	7 825 642	6 333+ 63 8 1059 9A7
6 1283 1868	H= 5, K= 1	0 3922 3543	0 1151 1349		4 . 514 501	3 1792 1765	8 344 320	5 1235 1165	0 744.0 181		
H= 4, K= 0	1 3451 3272	2829 2904	4 1312 1333	He 19, Ke 3	6 1560 1555 8 270+ 281	5 1134 1021	H= 10, K= 6		2 1520 1548		N= 12, K= 10
0 2716 2656	3 1879 1890 5 1950 2005	6 498 654	6 310+ 271	1 1253 1355			0 2935 2884	Ma 7, Ka 7	4 273 • 124	1 1231 1154	0 323+ 5
4 1789 1961	7 928 1004		0 000 741	5 862 935	N= 10, K= 4	H= 13, K+ 5	2 534 562	1 2235 2173	8 293 • 58	5 1223 1145	6 1008 1039
6 1202 172V	H= 7. K= 1	NN 4, EN 2	H= 24. K= 2	7 1046 1096	0 2581 2667	1 1071 1147	6 418 454	5 965 919	H= 8, K= 8	N= 9. K= 9	8 344+ 42
H= 6. X= 0		0 1871 1638	0 309+ 166	H= 21, x= 3	4 1845 1898	5 996 912	8 987 883	7 755 657			H= 14, K= 10
2 3178 3037	3 1879 1905	4 1014 1029	2 1081 1184 4 318+ 179	1 1335 1475	6 274+ 275	7 987 969	H= 12, K= 6	N= 9, K= 7	2 578 400	3 1468 1506	0 1401 1401
6 2040 2155	5 2219 2216	6 2017 2059	6 1013 1129	3 870 993	0 000 000	H= 15, K= 5	0 402 37	1 1204 1233	4 1588 1539	5 530 397	4 655 1025
H= 8, K= 0	1 .43 030	5 332 440	X= 26, 5= 2	5 904 978	H= 18, K= 4	1 1740 1008	2 2319 2271	3 1094 1212	8 934 791	'He 11, 24 0	6 348• 92
0 4342 4267	H= 9, E= 1	H= 6, K= 2	3 330- 170		0 271+ 298	3 1175 1157	6 1216 4187	5 1091 1030	NO 10. CO 8	1 1333 +320	H= 16, K= 10
4 2513 2527	1 2121 2091	0 2277 2251	2 3240 434	N= 23, K= 3	2 2330 2495	5 1667 1655	8 284+ 114			3 975 967	0 342+ 289
0 10/4 1/40	5 1169 1160	2 365 373	X= 1, K= 3	1 1335 1405	6 1018 989		H= 14, K= 6	/	2 2038 2015	5 947 1024	4 350+ 211
N= 10, K= 0	7 1264 1324	6 167+ 183	1 3748 3315	5 676 570	0 3000 138	· · · · · ·	0 2432 2431	1 1160 1141	4 597 512	H= 13, E= Q	H= 1, K= 11
2 3371 3386	H= 11, K= 1	6 1001 1137	5 993 1001	N= 0. K= 4	H= 20, K= 4	1 1036 1084	2 553 530	5 1144 1081	8 307+ 276	1 1289 1257	1 1403 1414
0 1545 1541	1 1133 1150	H= 8, K= 2	7 671 817		0 1594 1779	5 1074 1074	6 541 465	7 1031 1008	NN 12. EN 8	3 949 841	3 1080 1116
H= 12, K= 0	3 1701 1750	0 1750 1541	H= 3. K= 3	4 2154 2083	2 200 244	7 870 901	8 972 936	K= 13, K= 7		5 1127 1142	7 566 487
0 4607 4750	7 1063 1038	2 2725 2766	1 1470 1087	8 1700 1718	6 444 295	H= 19, K= 5	H= 16. K= 6	1 1028 1078	2 290+ 75	H# 15, E# 9	** * ** **
4 2016 1997 8 807 718	No. 13. 70. 1	6 1577 1555	3 2053 1885	H= 2, K= 4	0 /02 014	1 695 714	6 415 316	3 971 989	4 1872 1889	1 906 1048	
	N- 137 K- 1	8 441 485	5 1832 1858	0 706 592	H# 22, K# 4	3 1209 1246	2 1513 1623	7 790 812	8 877 725	5 332 • 351	1 701 667
14, K- U	3 1462 1513	N= 10, K= 2		2 2641 2521	0 304+ 170	7 836 818	6 1273 1298	#= 15, K= 7	H. 14, C. 8	HB 17. ER 0	5 732 707
2 2778 2919	5 1517 1508	0 2800 2872		6 2069 2014	4 313+ 7	H= 21. K= 5	8 308 • 271				1 112 081
	/ *// 1004	4 2567 2507	3 2836 2512	8 306 294	6 1092 1249	1 1180 1734	H= 18, X= 6	3 1042 975	2 1485 1493	3 728 810	H= 5, K= 17
HE 16, KE 0	H= 15, K= 1	6 622 642	5 1458 1373	H= 4, K= 4	•	3 519 781	0 1159 1112	5 886 854	4 513 531		1 692 687
0 2307 2203	1 1728 1792	8 1187 1202	/ 1609 16/1	0 3741 3007	H= 24, E= 4	5 1129 1177	2 296* 116		8 324= 313	N= 0. K= 10	5 840 890
8 1498 1591	5 931 944	H= 12, K= 2	H= 7, K= 3	2 1067 885	2 323+ 113		6 312+ 162		H= 16, E= 8	2 1341 1344	N= 7. K= 11
NY 18. EM 0	7 1048 1108	0 396 449	1 1911 1748	6 359 348	• ••• 103•	"" 23, t u 5	8 1065 1056	1 1546 1706	0 770 740		1 936 818
	N= 17, K= 1	4 213+ 247	5 1103 1065 5 1277 1204	8 1173 1067	H# 1, K# 5	1 667 1038	H= 20, K= -6	5 818 803	2 311+ 256	N 2, K 10	5 963 885
6 1444 1430	1 1956 2173	6 2138 2119	7 1401 1377	H= 6, K= 4	1 1926 1826	H= 0, K= 6	0 452 522	7 369 530	4 1216 1238 6 327+ 271	0 1491 1506	N= 9, E= 11
	3 1326 1368	6 6000 87	H= 9, K= 3	0 719 658	5 1520 1386	2 3575 3260	2 1277 1335	H= 19, K= 7	8 876 714	4 1172 1136	1 757 811
- 20, k = 0	7 670 675	H= 14, K= 2	1 2350 2318	2 4201 4061	7 1239 1284	6 1687 1645	6 868 930	1 1393 1575	H= 18, K= 8	6 316+ 225 8 1056 1011	5 745 081
0 1676 1674	N= 10 FA 4	0 2143 2185	3 1333 1338	6 1524 1393	H= 3, K= 5	H= 2, K= 6	8 333• 269	3 1238 1143			#= 11, K= 11
8 999 1024	- IV. K- 1	6 1923 1999	5 1537 1543 7 939 856	8 214+ 213	1 2274 2107		H= 22, K= 6	7 335+ 587	4 330+ 106	H= 4, K= 10	1 1124 1111
H= 22, K= 0	1 1559 1730	6 564 636		H= 8, K= 4	3 1196 1096	2 1361 1322	0 1567 1913	H= 21, E= 7	6 1031 1127	0 445 367	
	3 1326 1369	8 1000 414	N= 11, N= 3	0 3276 3288	5 1968 1863 7 661 542	4 2028 1967	2 325+ 527			4 309+ 274	Nº 0, Fº 12
6 1014 1174	/ 00/ 558	H= 16, K= 2	1 3259 3166	2 837 830		8 1191 1042	6 340+ 216	1 009 800	H= 20. K= 8	6 888 790 8 3269 137	0 1026 956
N= 24. K= 0	H= 21, K= 1	0 656 699	5 1182 1192	6 412 385	··· ›, ·· ›	H= 4, K= 6	N= 24. K= 6	H= 0. K= 8	0 1078 1230		
	1 1035 1135	4 530 546	7 924 913	8 1199 1121	1 1279 1272	0 344 377		0 2017 2076	- 120- 1323	Nº 0, Kº 10	W= 2, K= 12
4 1306 1500	3 1367 1527 5 611 584	6 1288 1231	H# 13, K= 3	H= 10, K= 4	5 1455 1294	2 2071 1971	2 1360 1500	8 757 544	H= 1, K= 0	0 2591 2475	0 525 529
	7. 1004 1030	-	1 2646 2600	0 504 579	7 1134 1124	4 360 400 6 1640 1490	H= 1, K= 7	v., 3 v., a	1 1481 1422	4 1536 1435	4 347• 482
	H= 23, K= 1	H= 18, K= 2	3 2168 2163	2 1995 1990	X= 7, K= 5	8 283+ 325	1 1150 1136		5 1167 1162	6 322+ 77 8 658 631	6 750 864
2 1003 1171	1 704 745	0 1830 1955	7 901 910	6 1871 1833	1 1746 1681	X= 6, K= 6	5 1255 1193 5 1066 973	0 1007 1027	7 730 673		H= 4. E= 12
H# 1, K# 1	3 953 1102	4 1991 2129	H= 15, K= 3	8 271+ 310	3 2330 2248	0 2070 1874	7 1174 1147	4 709 719	н» 3. к» 9	0, K= 10	0 1299 1311
1 2450 2269	> 935 1009	6 275+ 249	1	H= 12, K= 4	7 1367 1365	2 460 429	H= 3, K= 7	6 T>37 1386 8 443 284	1 1475 1343	0 544 363	4 1223 1174
3 1189 2208	H= 25, K= 1		3 900 882	0 1882 1917	H= 9, K= 5	4 1972 1902 6 452 217	1 1867 1773		3 1552 1504	4 318+ 275	MB 6, KB 12
7 545 1059	1 836 1006	H= 20, K= 2	5 1258 1229	2 361 206		8 1517 1667	3 1489 1433		7 919 853	o 980 1061 8 333+ 144	0 344+ 244
H= 3, K= 1	KB 0. FB 2	0 675 675		6 244+ 100	3 1354 1333	N* 8. X* 6	5 1475 1370 7 1072 958	C 1893 1802	No. 6. 60. 0		4 352 194
1 3804 3604		4 542 602	- 17, K# 3	8 1454 1437	5 1931 1884	0 370 340		4 2113 2070		10, 10 10	N= 8, K= 12
3 2790 3029	6 2208 2180	6 1204 1289 8 303+ 412	1 1125 1204	H= 16, K= 6		2 2406 2318)/ x- /	8 1091 1104	1 1249 1227 3 937 861	0 1201 1175	0 961 1060

4(b)] vary from 2.77 to 3.03 Å. There are no other barium oxygen contacts below 3.2 Å, and for the purpose of applying Pauling's rules (see below) these longer contacts are assumed not to represent bonds. The Al-O tetrahedron is somewhat irregular, with the Al-O(2) bond significantly longer than the rest. A possible explanation of this is that this oxygen atom forms bonds to two barium atoms, and that this weakens the Al-O linkage. This can be illustrated by considering the different ways in which Pauling's rules can be applied to the structure:

(a) Assume that Al-O has bond order 1, and Al-OH bond order $\frac{1}{2}$. This leads to the following picture:



which is unsatisfactory on several counts, although it does explain why Ba-O(3) is shorter than either of the Ba-O(2) distances. However the enormous variation in bond order for the variuos Ba-O contacts is not supported by the distances listed in Table 4.

(b) Assume that all Al-O bonds are of equal order *i.e.* $\frac{3}{4}$. The picture is then:



This is again unsatisfactory, since it would suggest that Ba-O(1) should be the shortest contact.

(c) Assume that all Ba–O bonds are equal, *i.e.* of order $\frac{1}{4}$. This leads to:



and this is in many ways the most satisfactory picture. It explains the variation of Al–O bond lengths very neatly, although it does not account for the variations in Ba–O contacts.

Table 4. Coordination polyhedra in y-Ba[AlO(OH)₂],

Figures in brackets give estimated standard deviations corresponding to the least significant digit.

Rond

(a) Al-O tetrahedron

		distances	
Al- Al-	O(1) (bridge) O(1')(bridge)	1·74 (1) Å 1·73 (1)	
Al-	O(2) (hydroxyl)	1.84 (1)	
Al-	O(3) (hydroxyl)	1.75 (1)	
	Bond angles		Bond angles
O(1)-Al-O(1')	105·8 (5)°	O(1')-Al-O(2)	108·3 (6)°
O(1)-Al-O(2)	108-9 (6)	O(1')-Al-O(3)	111.8 (5)
O(1)-Al-O(3)	115-1 (6)	O(2)-Al-O(3)	106.8 (6)

(b) Ba-O coordination

Since the barium atoms lie on twofold axes, each contact occurs twice.

Ba-O(1')	2·930 (9) Å
Ba-O(2)	2.84 (2)
Ba-O(2')	3.03 (2)
Ba-O(3')	2 77 (1)

All of the above represent over-simplifications in one way or another, and the true situation presumably lies somewhere between these three assignments.

It is interesting to compare this structure with that of α -Ba[AlO(OH)₂]₂ (Ahmed & Dent Glasser, 1970), which also contains [AlO(OH)₂⁻]_{∞} chains. In the α modification the chains are kinked into an almost helical configuration, they run parallel to one another, and there is hydrogen bonding between the chains. It may well be that the γ -modification is difficult to prepare because the arrangement of chains in a crisscross pattern is inherently less favoured than arrangement in a parallel one; the absence of hydrogen bonding and distortion of the Al–O tetrahedra may also betoken strain in the structure.

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